

Phase Study of the $(\text{La}, \text{Ln}, \text{Ce})_2\text{CuO}_4$ ($\text{Ln} = \text{Tb}, \text{Dy}, \text{and Ho}$) System

Y. K. TAO,* Y. Y. SUN, J. PAREDES, P. H. HOR, AND C. W. CHU

Texas Center for Superconductivity, University of Houston, Houston, Texas 77024-5506

Communicated by J. M. Honig, May 31, 1989

We studied the formation of the $(\text{La}, \text{Ln})_2\text{CuO}_4$ T' -structure with a small amount of Ce-doping. The results show that LaLnCuO_4 has a T' -structure for $\text{Ln} = \text{Tb}$. The addition of Ce is necessary to form the T' -structure for $\text{Ln} = \text{Dy}$ and Ho . We also report the T^* -structure $\text{La}_{1.1}\text{Dy}_{0.9}\text{CuO}_4$. © 1989 Academic Press, Inc.

The lanthanide cuprate oxide of general formula Ln_2CuO_4 has been reported (1-5) to form two types of structures: For $\text{Ln} = \text{La}$, it forms a K_2NiF_4 structure; for $\text{Ln} = \text{Pr}, \text{Nd}, \text{Sm}, \text{Eu}, \text{and Gd}$, it forms a Nd_2CuO_4 structure (referred to as the T' -structure hereafter). The ternary oxides $\text{LnLn}'\text{CuO}_4$ also form a T' -structure (6) when (i) $\text{Ln} = \text{La}$; $\text{Ln}' = \text{Pr}, \text{Nd}, \text{Sm}, \text{Eu}, \text{Gd}$, but not Tb and Dy , and (ii) when $\text{Ln} = \text{Pr}, \text{Nd}, \text{Sm}, \text{Eu}$; $\text{Ln}' = \text{Nd}, \text{Sm}, \text{Eu}, \text{Gd}, \text{Tb}, \text{and Dy}$. Among all these oxides, only La_2CuO_4 shows a metallic behavior, whereas all others are semiconductors (7). The recent discovery of superconductivity in $\text{Nd}_{2-x}\text{Ce}_x\text{CuO}_4$ has attracted great attention because of the electron-character of the charge carriers in these compounds as evident from the Hall and Seebeck measurements (8). In a study of superconductivity in the $\text{LaLn}_{1-x}\text{Ce}_x\text{CuO}_4$ system (9) ($\text{Ln} = \text{rare-earth element}$), $\text{LaTb}_{0.85}\text{Ce}_{0.15}\text{CuO}_4$ and $\text{LaDy}_{0.85}$

$\text{Ce}_{0.15}\text{CuO}_4$ were reported to form the T' -structure. The formation of the T' -structure in these two cases had not been reported previously and was further studied in this work.

Most samples were prepared as follows: An appropriate amount of rare-earth oxides and CuO were mixed and fired at 950°C for 8-16 hr. The powders thus obtained were pelletized into 5-mm-diameter * 1.5-mm-thick disks and fired at 1050°C (referred to as second-firing temperature) for another 8-16 hr, then air quenched. Some samples had a different second-firing temperature, which will be specified in a later publication.

In contrast to Nedil'ko's results (6), the samples with nominal compositions LaTbCuO_4 and $\text{LaTb}_{0.85}\text{Ce}_{0.15}\text{CuO}_4$ show a pure T' -structure as indicated by the X-ray diffraction pattern. The synthesis route of Nedil'ko was different from ours, which might account for the difference in results. After annealing in flowing Ar at 900°C for 16 hr,

* To whom correspondence should be addressed.

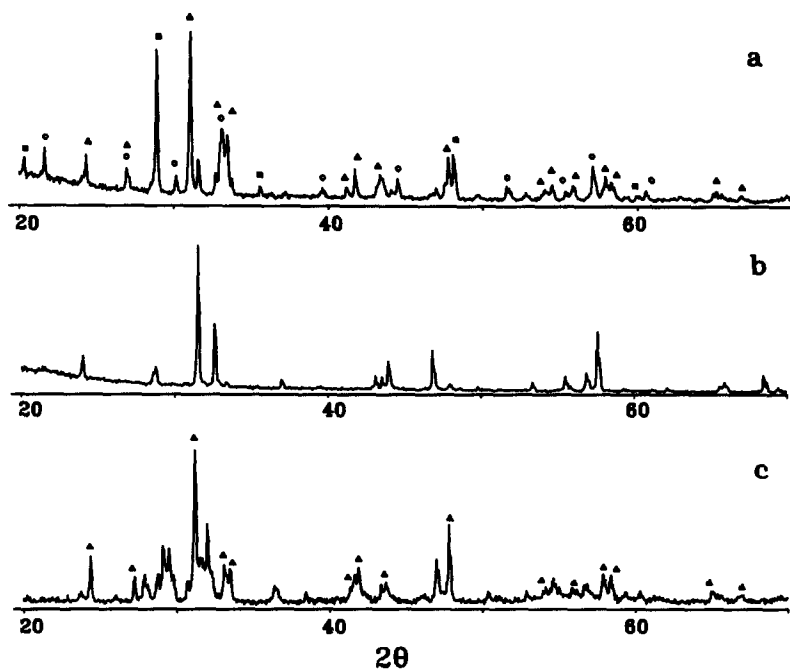


FIG. 1. The X-ray diffraction patterns of $\text{La}_{1.1}\text{Dy}_{0.9}\text{CuO}_4$ samples with second-firing temperatures (a) 950°C (\blacktriangle , La_2CuO_4 ; \circ , $\text{Dy}_2\text{Cu}_2\text{O}_5$; and \blacksquare , Dy_2O_3), (b) 1050°C , and (c) 1200°C .

both samples show semiconducting behavior from room temperature down to liquid helium temperature.

The same preparation technique was used to form LaDyCuO_4 . However, a mixed-phase sample was obtained. The major feature in the X-ray pattern remains quite simple and matches well with the T^* -structure reported by Takayama-Muramachi *et al.* (10). (The T^* -structure consists of alternative stackings of K_2NiF_4 -type slabs and T' -type slabs.) By adjusting the La:Dy ratio, the purest T^* -phase was found in the $\text{La}_{1.1}\text{Dy}_{0.9}\text{CuO}_4$ sample. The X-ray diffraction patterns of $\text{La}_{1.1}\text{Dy}_{0.9}\text{CuO}_4$ samples fired at different second-firing temperatures are shown in Fig. 1. This shows that this T^* -phase forms only over a limited temperature range. With a small amount of Ce substitution for Dy, the phase immediately changes from the T^* - to the T' -structure. Figure 2 shows the X-ray diffraction

patterns of the T^* -structure $\text{La}_{1.1}\text{Dy}_{0.9}\text{CuO}_4$ and the T' -structure $\text{LaDy}_{0.85}\text{Ce}_{0.15}\text{CuO}_4$. After annealing in flowing Ar at 900°C for 16 hr, $\text{LaDy}_{0.85}\text{Ce}_{0.15}\text{CuO}_4$ exhibits semiconducting behavior from room temperature down to liquid helium temperature.

By optical microscopy, the product of nominal composition LaHoCuO_4 appears to be a cluster of black and green particles. La_2CuO_4 , $\text{Ho}_2\text{Cu}_2\text{O}_5$, and Ho_2O_3 were identified in the X-ray diffraction pattern. With a small amount of Ce substitution for Ho, the T' -phase forms. The X-ray diffraction pattern of $\text{LaHo}_{0.85}\text{Ce}_{0.15}\text{CuO}_4$ shows that the major phase consists of the T' -structure.

In summary, we found that the T' -phase exists in $(\text{La}, \text{Ln})_2\text{CuO}_4$ with Ln extended beyond Gd to Ho with the presence of a small amount of Ce-substitution. $\text{LaTb}_{0.85}\text{Ce}_{0.15}\text{CuO}_4$ and $\text{LaDy}_{0.85}\text{Ce}_{0.15}\text{CuO}_4$ exhibit semiconducting behavior from room tem-

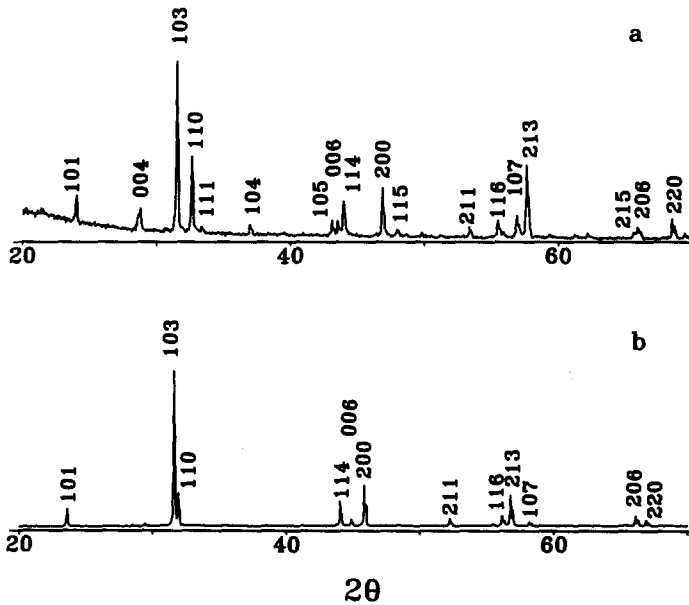


FIG. 2. The X-ray diffraction patterns of (a) T^* -structure $\text{La}_{1.1}\text{Dy}_{0.9}\text{CuO}_4$ and (b) T' -structure $\text{LaDy}_{0.85}\text{Ce}_{0.15}\text{CuO}_4$.

perature down to liquid helium temperature. The possible connection between the formation of the T' -structure by addition of Ce and superconductivity induced by Ce-doping in the T' -structure are being investigated. By varying the La to Dy ratio, a pure T^* -structure compound $\text{La}_{1.1}\text{Dy}_{0.9}\text{CuO}_4$ was also obtained.

Acknowledgments

This work is supported in part by the NSF Low Temperature Physics Program Grant No. DMR 86-126539, DARPA Grant No. MDA 972-88-G-002, Texas Center for Superconductivity at the University of Houston, and The T.L.L. Temple Foundation.

References

1. YU D. TRET'YAKOV, A. R. KAUL, AND N. V. MAKUKHIN, *J. Solid State Chem.* **17**, 183 (1976).
2. L. M. LONGO AND P. M. RACCAH, *J. Solid State Chem.* **6**, 526 (1973).
3. VON HK MULLER-BUSCHBAUM AND V. WOLLSCHLAGER, *Z. Anorg. Allg. Chem.* **76**, 414 (1975).
4. I. S. SHAPLYGIN, B. G. KAKHAN, AND V. B. LAZAREV, *Zh. Neorg. Khim.* **24**, 1478 (1979).
5. S. A. NEDIL'KO, M. N. ERMAKOVA, N. P. ZYRYANOVA, R. D. VASYAGINA, AND S. G. KRASIL'NIKOVA, *Izv. Akad. Nauk. SSSR Neorg. Mater.* **16**, 1126 (1980).
6. S. A. NEDIL'KO, *Zh. Neorg. Khim.* **27**, 1130 (1982).
7. P. GANGULY AND C. N. R. RAO, *Mater. Res. Bull.* **8**, 405 (1973).
8. H. TAKAGI, S. UCHIDA, AND Y. TOKURA, *Phys. Rev. Lett.* **62**, 1197 (1989).
9. M. BONVALOT, Y. K. TAO, Y. Y. SUN, Q. XIANG, Z. J. HUANG, G. LIN, J. BECHTOLD, R. L. MENG, P. H. HOR, AND C. W. CHU, to be published.
10. E. TAKAYAMA-MUROMACHI, Y. MATSUI, Y. UCHIDA, F. IZUMI, M. ONODA, AND K. KATO, *Japan. J. Appl. Phys.* **27**, L2283 (1988).